Access DB# 111446

#### SEARCH REQUEST FORM

#### Scientific and Technical Information Center

Requester's Full Name: FAIC COMPTON	Examiner #: 77/32 Date: 1/6/04
Requester's Full Name: Enla ComptoN  Art Unit: 3726 Phone Number 30 5- 62 40	Serial Number: 10/069, 384
Mail Box and Bldg/Room Location: CP2 FA / 3 Resu	Ilts Format Preferred (circle): PAPER DISK E-MAIL
If more than one search is submitted, please prioritiz	
Please provide a detailed statement of the search topic, and describe a Include the elected species or structures, keywords, synonyms, acron utility of the invention. Define any terms that may have a special me known. Please attach a copy of the cover sheet, pertinent claims, and	yms, and registry numbers, and combine with the concept or aning. Give examples or relevant citations, authors, etc, if abstract.
Title of Invention: NITINOL BALL STAINS E.  Inventors (please provide full names): JULIEN GE	LEMENT AND PROCESS FOR MAKING
Inventors (please provide full names):	TAALD
	-
Earliest Priority Filing Date: 8 / 19 / / 99	29
*For Sequence Searches Only* Please include all pertinent information (pappropriate serial number.	المستحدد الم
	( START MEMORY ACTORY)
BALL SEARING + NITINOL	(SAME MEMOR)
ROLLER BEARING	
RACE ELEME	ent.
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Date Searcher Picked Up: 10/04 Bibliographic	Dr.Link
12	Lexis/Nexis
Searcher Prep & Review Time: 140 41N Fulltext	Sequence Systems
Clerical Prep Time: Patent Family	WWW/Internet V
Online Time: 167 MINOther	Other (specify)

Other (specify)\_



# STIC Search Report

### STIC Database Tracking Number: 111446

TO: Eric Compton Location: cp2 5a13

Art Unit: 3726

Tuesday, January 06, 2004

Case Serial Number: 10/069384

From: Emory Damron Location: EIC 3700

CP2-2C08

Phone: 305-8587

Emory.Damron@uspto.gov

#### Search Notes

Dear Eric,

Please find below an inventor search in the bibliographic and full-text foreign patent files, as well as keyword searches in the patent and non-patent literature files, both bibliographic and full text.

References of potential pertinence have been tagged, but please review all the packets in case you like something I didn't.

<u>In addition</u> to searching on Dialog, I also searched Google, EPO/JPO/Derwent, and STN.

Please contact me if I can refocus or expand any aspect of this case.

Happy New Year!

Sincerely, Emory Damron Technical Information Specialist EIC 3700, US Patent & Trademark Office Phone: (703) 305-8587/ Fax: (703) 306-5915

Emory.damron@uspto.gov



## EIC 3700

Questions about the scope or the results of the search? Contact the EIC searcher or contact:

John Sims, EIC 3700 Team Leader 308-4836, CP2-2C08

Voluntary Results Feedback Form
> I am an examiner in Workgroup: Example: 3730
> Relevant prior art found, search results used as follows:
☐ 102 rejection
☐ 103 rejection
☐ Cited as being of interest.
☐ Helped examiner better understand the invention.
Helped examiner better understand the state of the art in their technology.
Types of relevant prior art found:
☐ Foreign Patent(s)
<ul> <li>Non-Patent Literature         <ul> <li>(journal articles, conference proceedings, new product announcements etc.)</li> </ul> </li> </ul>
> Relevant prior art not found:
☐ Results verified the lack of relevant prior art (helped determine patentability).
Results were not useful in determining patentability or understanding the invention.
Comments:

Drop off or send completed forms to STIC/EIC3700 CP2 2C08



(FILE 'HOME' ENTERED AT 14:21:38 ON 06 JAN 2004)

FILE 'A	LUMINIUM,	HCAPLUS,	MATBUS,	METADEX,	EMA,	AEROSPACE,	CONFSCI,
RUSSCI'	ENTERED	AT 14:22:	30 ON 06	JAN 2004			

	KOSSCI, FUIFK	ED AT 14:22:30 ON 06 JAN 2004
L1	36435 S	NITINOL OR SHAPE MEMORY OR SHAPEMEMORY OR NITI OR NI TI OR TI
L2	42379 S	SMA OR SMM OR SME OR MARTEN? (5N) AUSTEN? OR TINI OR TI NI
L3	69592 S	L1-L2
L4	7800 S	BALL BEARING? OR BEARING BALL? OR RACE BEARING? OR BEARING RA
L5	5857 S	ROLLER BEARING? OR NEEDLE BEARING? OR BEARING CAGE? OR CAGE B
L6	12648 S	L4-L5
L7	96 S	L3 AND L6
L8	90 S	L7 AND PY<=2000
L9	79 DU	P REM L8 (11 DUPLICATES REMOVED)

L9 ANSWER 4 OF 79 AEROSPACE COPYRIGHT 2004 CSA on STN

ACCESSION NUMBER: 2002:007231 AEROSPACE

DOCUMENT NUMBER: A98-36400

TITLE: Shape memory wire as a drive mechanism

for a Fourier transform spectrometer Brasunas, John, (NASA, Goddard Space Flight Center,

AUTHOR(S): Brasunas, John Greenbelt, MD)

SOURCE: Optical Engineering, (Jun 1998) vol. 37, no.

636908, pp. 1882, 1883. Refs: 4. Available from: Aeroplus

Dispatch.

ISSN: 0091-3286

PUB. COUNTRY: United States

DOCUMENT TYPE: LANGUAGE:

Journal English

AB This paper examines the use of shape memory wire as a potentially much lighter and cheaper alternative to the voice-coil motor of a Fourier transform spectrometer (FTS). For instance, Flexinol , which is manufactured by Dynalloy, Inc., of Irvine, CA, is a shape (length) memory alloy actuator wire made of nickel and titanium. When prestretched at room temperature and then electrically driven (heated), it contracts. The alloy changes its internal structure at a transition temperature (typically set at 70-90 C). The contraction during heating may be as great as a few percent. The wire can then be stretched again when cooled, and this can be repeated many times. The motion (stroke) of the memory wire is set by the stress one uses to stretch the below-transition wire. For 10,000-psi loading, the stroke is about a four-percent stretch. For 0.010-in.-diameter Flexinol, the maximum pull is 930 g. In comparison, the voice-coil motor may supply a force up to one lb. A servo-controlled FTS experimental setup using 30 cm of 0.010-in.-diameter Flexinol as a shape memory wire drive is demonstrated. The results show that for the same FTS servo network, tachometer, and ball-bearing slide, the memory wire drive has a speed variation that is about a factor of two worse than the voice-coil drive. This is, nevertheless, a respectable performance, since a 9.5-percent speed variation for such a low speed as 55-micron/s is still useful for Fourier transform spectroscopy.

TI Shape memory wire as a drive mechanism for a Fourier transform spectrometer

SO Optical Engineering, (Jun 1998) vol. 37, no. 636908, pp. 1882, 1883. Refs: 4. Available from: Aeroplus Dispatch. ISSN: 0091-3286

This paper examines the use of **shape memory** wire as a potentially much lighter and cheaper alternative to the voice-coil motor of a Fourier transform spectrometer (FTS). For instance, **Flexinol**, which is manufactured by Dynalloy, Inc., of Irvine, CA, is a shape (length) memory alloy actuator wire made of nickel. . . stress one uses to stretch the below-transition wire. For 10,000-psi loading, the stroke is about a four-percent stretch. For 0.010-in.-diameter **Flexinol**, the maximum pull is 930 g. In comparison, the voice-coil motor may supply a force up to one lb. A servo-controlled FTS experimental setup using 30 cm of 0.010-in.-diameter **Flexinol** as a **shape** memory wire drive is demonstrated. The results show that for the same FTS servo network, tachometer, and **ball-bearing** slide, the memory wire drive has a speed variation that is about a factor of two worse than the voice-coil. .

\*\*SHAPE MEMORY ALLOYS; \*FOURIER TRANSFORM SPECTROMETERS; \*ELECTRIC WIRE; TACHOMETERS; SERVOCONTROL

L9 ANSWER 17 OF 79 HCAPLUS COPYRIGHT 2004 ACS on STN DUPLICATE 2

ACCESSION NUMBER: 1991:86383 HCAPLUS

DOCUMENT NUMBER: 114:86383

TITLE: Application of isothermal martensite in GCr15

ball-bearing steel

AUTHOR(S): Xu, Zuyao; Chen, Yexin; Chen, Weiye

CORPORATE SOURCE: Shanghai Jiaotong Univ., Shanghai, Peop. Rep. China

SOURCE: Gangtie (1990), 25(6), 47-50 CODEN: KATIAR; ISSN: 0449-749X

DOCUMENT TYPE: Journal LANGUAGE: Chinese

AB Isothermal martensite promotes stabilization of retained austenite in the quenched GCr15 ball-bearing steel upon cooling to subzero temperature, cycling of compressive stress, and aging at room temperature Retained austenite was induced to transform to martensite under stress cycling to increase the contact fatigue life. Isothermal martensite stabilizes retained austenite and results in beneficial transformation-induced plasticity. The contact fatigue life of steel parts isothermally treated with a few percentages of isothermal martensite is 36% higher than that after subzero treatment and 18% higher than that treated by normal quenching and tempering. The dimensional stability of specimen treated isothermally is higher than that of specimens treated by normal quenching and tempering by 34%. The accommodation deformation induced by the isothermal martensite transformation leads to the mech. stabilization of retained austenite.

- TI Application of isothermal martensite in GCr15 ball-bearing steel
- SO Gangtie (1990), 25(6), 47-50 CODEN: KATIAR; ISSN: 0449-749X
- Isothermal martensite promotes stabilization of retained AB austenite in the quenched GCr15 ball-bearing steel upon cooling to subzero temperature, cycling of compressive stress, and aging at room temperature Retained austenite was induced to transform to martensite under stress cycling to increase the contact fatique life. Isothermal martensite stabilizes retained austenite and results in beneficial transformation-induced plasticity. The contact fatique life of steel parts isothermally treated with a few percentages of isothermal martensite is 36% higher than that after subzero treatment and 18% higher than that treated by normal quenching and tempering. The dimensional stability of specimen treated isothermally is higher than that of specimens treated by normal quenching and tempering by 34%. The accommodation deformation induced by the isothermal martensite transformation leads to the mech. stabilization of retained austenite.

L9 ANSWER 42 OF 79 METADEX COPYRIGHT 2004 CSA on STN

ACCESSION NUMBER: 1986(6):12-732 METADEX

TITLE: The Influence of Cyclic Tempering on the Structure of

Ball-Bearing Steels.

AUTHOR: Ignat, C.

SOURCE: Cercet. Metal. (1983) 24, 335-340

DOCUMENT TYPE: Journal LANGUAGE: Romanian

AB The effect of cyclic tempering on the relative amounts of residual austenite, martensite and carbide is investigated in ball-bearing steels. The carbide size, density and distribution variation by groups of dimensions is also reported. A thorough analysis of the results together with optic and electronic microscopy based observations enable one to conclude that cyclic tempering on the above-mentioned steels is a feasible technology for increasing size stabilization. 4 reference-AA

- TI The Influence of Cyclic Tempering on the Structure of Ball-Bearing Steels.
- SO Cercet. Metal. (1983) 24, 335-340
- AB The effect of cyclic tempering on the relative amounts of residual austenite, martensite and carbide is investigated in ball-bearing steels. The carbide size, density and distribution variation by groups of dimensions is also reported. A thorough analysis of the. . .

L9 ANSWER 44 OF 79 METADEX COPYRIGHT 2004 CSA on STN

ACCESSION NUMBER: 1983(10):61-616 METADEX

TITLE: System for Train Accident Reduction-DOT STAR.

AUTHOR: Richardson, J.J.
NUMBER OF REPORT: NBS Spec. Publ. 652

SOURCE: National Bureau of Standards. Washington, D.C. 20234.

1983. 132-145. Accession Number: 83(10):72-537 Conference: Damage Prevention in the Transportation Environment, Gaithersburg, Md., 21-23 Oct. 1981

DOCUMENT TYPE: Conference; Report

LANGUAGE: English

AB The Dept. of Transportation-System for Train Accident Reduction (DOT STAR) study at the Naval Surface Weapons Center (NSWC) is developing a prototype system to help reduce the number of train accidents. NSWC has taken military technology and applied it to develop an on-train anti-derailment system. This system can sense a local derailment or a hot bearing. Upon sensing these conditions, the system automatically applies emergency braking. Exploratory development hardware of the journal bearing thermal sensor successfully completed > 100 000 miles of travel. A roller bearing thermal sensor has also been designed. NITINOL is the key component used to sense overheated bearings. The derailment detector uses the impact of a sensor foot with the rail head to sense a local derailment. Upon generation of a hot box or a local derailment, the sensors initiate a thermal pulse battery. The electrical pulse activates the air valve which applies the train's brakes.-AA

NR NBS Spec. Publ. 652

SO National Bureau of Standards. Washington, D.C. 20234. 1983. 132-145. Accession Number: 83(10):72-537 Conference: Damage Prevention in the Transportation Environment, Gaithersburg, Md., 21-23 Oct. 1981

AB. . . emergency braking. Exploratory development hardware of the journal bearing thermal sensor successfully completed > 100 000 miles of travel. A roller bearing thermal sensor has also been designed.

NITINOL is the key component used to sense overheated bearings.

The derailment detector uses the impact of a sensor foot with. . .

CT Railroad cars; Roller bearings: Service life; Journal bearings: Service life; Shape memory; Safety; Sensors; Fatigue failure; Temperature measurement

L9 ANSWER 64 OF 79 HCAPLUS COPYRIGHT 2004 ACS on STN

ACCESSION NUMBER: 1967:31211 HCAPLUS

DOCUMENT NUMBER: 66:31211

TITLE: Suitability of high-temperature resistant

nickel-chromium alloys for dies for the extrusion of

steel

AUTHOR(S): SOURCE:

Jahn, Thomas; Gaertner, Gerhard Neue Huette (1966), 11(11), 680-4 CODEN: NEUHAM; ISSN: 0028-3207

DOCUMENT TYPE: Journal LANGUAGE: German

The high temps. and pressures (100 kg./mm.2) involved in extruding hot steel cause severe tool and die wear. High hot tensile strength, yield strength, resistance to wear and tear, and ductility are important properties for dies. An alloy with 8-10% W was worn out after 20-60 extrusions. Expts. were conducted with the wrought alloy LW 2325 containing C 0.10, SiO 42, Mn 0.26, Cr 15.68, W 5.25, Mo 1.38, Al 2.14, Ti 1.58, and Fe 1.68%, S, P, Cu, Ce, and B traces, and the balance Ni, HRC (Rockwell C Hardness) 39 as received and 31 after solution treatment and aging; LW 2326 with C 0.18, Si 0.34, Mn 0.39, Cr 14.91, W 1.39, Al 4.35, and Fe 1.68%, S, P, Cu, Ce, B, and Sb traces, and the balance Ni with resp. HRC 30 and 18; the Ni-Cr cast alloy I (C 0.13, Mn 0.47, Cr 12.04, Mo 2.80, Ti 1.76, Fe 4.8%, and the balance Ni (HRC 39, as received); and cast alloy II (C 0.24, Mn 0.32, Cr 11.92, Mo 2.25, Ti 1.68, Fe 6.7%, and the balance Ni) with HRC 39, as received). The die had a sleeve of W alloy and an insert of Ni-Cr alloy which can be changed after wear. Ball bearing tubes were extruded with outside diameter 23.6 mm., inside diameter 14.2 mm.,

tubes were extruded with outside diameter 23.6 mm., inside diameter 14.2 mm. and length 2.8 m. Billets with 73 mm. outside diameter, 16 mm. inside diameter,

and 210 mm. length were used. The billets were preheated at 1100° in a salt bath. A 500 + 106 g. hydraulic extrusion press with a ram speed of 300 mm./sec. was considered. The change in cross section was 93%. The most reliable alloy was LW 2325 as received, which survived  $\leq$ 400 extrusions. Aged dies of LW 2325 were too soft and survived  $\leq$ 160 extrusions. Dies of 30 WCrV 34.11 survived  $\leq$ 13 extrusions, of aged LW 2326  $\leq$ 285, cast alloy I  $\leq$ 173, and cast alloy II  $\leq$ 88 extrusions. LW 2325 was also suitable for extruding small steel profiles.

SO Neue Huette (1966), 11(11), 680-4 CODEN: NEUHAM; ISSN: 0028-3207

AB The high temps. and pressures (100 kg./mm.2) involved in extruding hot steel cause severe tool and die wear. High hot tensile strength, yield strength, resistance to wear and tear, and ductility are important properties for dies. An alloy with 8-10% W was worn out after 20-60 extrusions. Expts. were conducted with the wrought alloy LW 2325 containing C 0.10, SiO 42, Mn 0.26, Cr 15.68, W 5.25, Mo 1.38, Al 2.14, Ti 1.58, and Fe 1.68%, S, P, Cu, Ce, and B traces, and the balance Ni, HRC (Rockwell C Hardness) 39 as received and 31 after solution treatment and aging; LW 2326 with C 0.18, Si 0.34, Mn 0.39, Cr 14.91, W 1.39, Al 4.35, and Fe 1.68%, S, P, Cu, Ce, B, and Sb traces, and the balance Ni with resp. HRC 30 and 18; the Ni-Cr cast alloy I (C 0.13, Mn 0.47, Cr 12.04, Mo 2.80, Ti 1.76, Fe 4.8%, and the balance Ni (HRC 39, as received); and cast alloy II (C 0.24, Mn 0.32, Cr 11.92, Mo 2.25, Ti 1.68, Fe 6.7%, and the balance Ni) with HRC 39, as received). The die had a sleeve of W alloy and an insert of Ni-Cr alloy which can be changed after wear. Ball bearing

tubes were extruded with outside diameter 23.6 mm., inside diameter 14.2 mm., and length 2.8 m. Billets with 73 mm. outside diameter, 16 mm. inside diameter,

and 210 mm. length were used. The billets were preheated at 1100° in a salt bath. A 500 + 106 g. hydraulic extrusion press with a ram speed of 300 mm./sec. was considered. The change in cross section was 93%. The most reliable alloy was LW 2325 as received, which survived  $\leq\!400$  extrusions. Aged dies of LW 2325 were too soft and survived

≤160 extrusions. Dies of 30 WCrV 34.11 survived ≤13 extrusions, of aged LW 2326 ≤285, cast alloy I ≤173, and cast alloy II ≤88 extrusions. LW 2325 was also suitable for extruding small steel profiles. DIES STEEL EXTRUSION; BALL BEARING TUBES; STEEL ST EXTRUSION DIES Tungsten alloys, containing IT (aluminum-chromium-iron-molybdenum-nickel-titaniumand aluminum-chromium-iron-nickel-, for dies for extrusion of steel) ΙŢ Molybdenum alloys, containing (aluminum-chromium-iron-nickel-titanium-tungstenand chromium-iron-nickel-titanium-, for dies for extrusion of steel) IT Iron alloys, containing (aluminum-chromium-molybdenum-nickel-titanium -tungsten- and aluminum-chromium-nickel-tungsten- and chromium-molybdenum-nickel-titanium-, for dies for extrusion of steel) ΙT Chromium alloys, containing (aluminum-iron-molybdenum-nickel-titanium -tungsten-, aluminum-iron-nickel-tungsten-, and iron-molybdenumnickel-titanium-, for dies for extrusion of steel) ΙT Aluminum alloys, containing

-tungsten-, and chromium-iron-nickel-tungsten-, for dies for extrusion

(chromium-iron-molybdenum-nickel-titanium

of steel)

L9 ANSWER 65 OF 79 HCAPLUS COPYRIGHT 2004 ACS on STN

ACCESSION NUMBER: 1966:25760 HCAPLUS

DOCUMENT NUMBER: 64:25760
ORIGINAL REFERENCE NO.: 64:4708a-c

TITLE: Producing compressive stress in the surface layer of

hardened steel

INVENTOR(S): Koistinen, Donald P. PATENT ASSIGNEE(S): General Motors Corp.

SOURCE: 5 pp.; Division of U.S. 3,117,041

DOCUMENT TYPE: Patent Unavailable

FAMILY ACC. NUM. COUNT: 1

PATENT INFORMATION:

AB Steel articles hardenable to their centers are hardened to leave their surfaces in compression to improve the fatigue resistance, by altering either the composition or the austenitizing temperature of the surface of the article

compared to the interior so that the transformation temperature of austenite to martensite is lower at the surface. Then in quenching, martensite is formed 1st in the interior where the resulting volume change occurs while the surface layer is plastic, and when the surface transforms later and expands, the interior is hard and resists further expansion, producing compressive stress at the surface. Steels containing 0.95-1.1% C and 1.3-1.6% Cr, commonly used for hardened ball bearings, is specially suitable for this process. Such a steel ball can be austenitized throughout at 1450°F., and subsequently heated briefly to 1900°F. so that only the surface layer is affected, before the entire ball is quenched in oil. At about 500°F. the interior transforms to hard martensite, while the surface layer does not transform until the ball temperature falls to 250°F. when compressive stress results from its expansion. The same result can be obtained with a single 30-min. austenitizing at 1575°F. in an atmospheric containing 5% NH3, when N diffuses into the surface layer to a depth of about 0.015 in. On quenching in oil, martensite forms at a lower temperature in the nitrided layer than in the interior. By treating with N for 5-10 hrs. at 975-1050°F., the layer stressed in compression on quenching was thickened to 0.045 in. Cr, Ni, or B can be applied to the surface and diffused inward to obtain the same effect as with N on hardening.

PI US 3216869 **19651109** 

PATENT NO. KIND DATE APPLICATION NO. DATE
PI US 3216869 19651109 US 19630429 <--

AB Steel articles hardenable to their centers are hardened to leave their surfaces in compression to improve the fatigue resistance, by altering either the composition or the austenitizing temperature of the surface of the article

compared to the interior so that the transformation temperature of austenite to martensite is lower at the surface. Then in quenching, martensite is formed 1st in the interior where the resulting volume change occurs while the surface layer is plastic, and when the surface transforms later and expands, the interior is hard and resists further expansion, producing compressive stress at the surface. Steels containing 0.95-1.1% C and 1.3-1.6% Cr, commonly used for hardened ball bearings, is specially suitable for this process. Such a steel ball can be austenitized throughout at 1450°F., and subsequently heated briefly to 1900°F. so that only the surface layer is affected, before the entire ball is quenched in oil. At about 500°F. the interior transforms to hard martensite, while the

surface layer does not transform until the ball temperature falls to  $250\,^\circ\mathrm{F}$ . when compressive stress results from its expansion. The same result can be obtained with a single 30-min. austenitizing at  $1575\,^\circ\mathrm{F}$ . in an atmospheric containing 5% NH3, when N diffuses into the surface layer to a depth of about 0.015 in. On quenching in oil, martensite forms at a lower temperature in the nitrided layer than in the interior. By treating with N for 5--10 hrs. at  $975\text{--}1050\,^\circ\mathrm{F}$ ., the layer stressed in compression on quenching was thickened to 0.045 in. Cr, Ni, or B can be applied to the surface and diffused inward to obtain the same effect as with N on hardening.

#### IT Bearings

(ball, with compressive surface stress)

ANSWER 68 OF 79 HCAPLUS COPYRIGHT 2004 ACS on STN

ACCESSION NUMBER: 1964:402543 HCAPLUS

DOCUMENT NUMBER:

61:2543

ORIGINAL REFERENCE NO.: 61:375f-h,376a

TITLE:

Complete transformation of austenite in hardened

high-carbon steel

INVENTOR(S): PATENT ASSIGNEE(S): Mantel, Edward R. General Motors Corp.

SOURCE: DOCUMENT TYPE:

3 pp. Patent

LANGUAGE:

Unavailable

PATENT INFORMATION:

PATENT NO. KIND DATE APPLICATION NO. DATE 19640428 US 19620223 -----19620223 <--

High-C Cr steel roller and ball bearings are hardened AB and tempered with dimensional stability, no retained austenite, and hardness maintained at 62-3 Rockwell C, by quenching in oil at 90-100°F. from 1750-1825°F., re-austenitizing at 1550-1600°F. and similarly quenching, immediately refrigerating 2 hrs. at -110°F., warming slowly, and tempering at 400-500°F., preferably for 4 hrs. at 400, 4 hrs. at 450, and 2 hrs. at 500°F. Double austenitizing and quenching give a finer structure, higher-C martensite, and greater strength, and with the cold treatment permit higher tempering, to transform all the retained austenite, without dimensional change or softening. When test bars 0.25 + 0.5 + 0.75 in. of steel containing C 1.01, Mn 0.34, Si 0.26, Ni 0.12, Cr 1.41, Mo 0.02, S 0.02, and P 0.008% were so treated, first 30  $\,$ min. at 1800°F., then 1 min in oil at 95°F., 30 min. at 1575°F., 1 min. in the oil again, and within 1 min. cooled to -110°F. for 2 hrs. and tempered 10 hrs. as stated above, the hardness was 62.5 Rockwell C, precision elastic limit by foil strain gage 88,000 lb./in.2 and tensile strength 314,000327,000 lb./in.2 Specimens conventionally heat treated and containing 2-3% retained austenite had 44,000 1b./in.2 elastic limit and 59-60 Rockwell C hardness.

PIUS 3131097 19640428

PΙ

PATENT NO. KIND DATE APPLICATION NO. DATE \_\_\_\_\_\_ 19640428 US 19620223 <--

AΒ High-C Cr steel roller and ball bearings are hardened and tempered with dimensional stability, no retained austenite, and hardness maintained at 62-3 Rockwell C, by quenching in oil at 90-100°F. from 1750-1825°F., re-austenitizing at 1550-1600°F. and similarly quenching, immediately refrigerating 2 hrs. at -110°F., warming slowly, and tempering at 400-500°F., preferably for 4 hrs. at 400, 4 hrs. at 450, and 2 hrs. at 500°F. Double austenitizing and quenching give a finer structure, higher-C martensite, and greater strength, and with the cold treatment permit higher tempering, to transform all the retained austenite, without dimensional change or softening. When test bars 0.25 + 0.5 + 0.75 in. of steel containing C 1.01, Mn 0.34, Si 0.26, Ni 0.12, Cr 1.41, Mo 0.02, S 0.02, and P 0.008% were so treated, first 30 min. at 1800°F., then 1 min in oil at 95°F., 30 min. at 1575°F., 1 min. in the oil again, and within 1 min. cooled to -110°F. for 2 hrs. and tempered 10 hrs. as stated above, the hardness was 62.5 Rockwell C, precision elastic limit by foil strain gage 88,000 lb./in.2 and tensile strength 314,000327,000 lb./in.2 Specimens conventionally heat treated and containing 2-3% retained austenite had 44,000 1b./in.2 elastic limit and 59-60 Rockwell C hardness.

ΙT 12173-93-2, Martensite

(formation, from residual austenite in steel for bearings)

ΙT 12244-31-4, Austenite (residual or retained,  $\mbox{martensite}$  formation from, in steel for bearings)

L Number	Hits	Search Text	DB	Time stamp
5	20	(shape adj memory) same (ball adj bearing\$	USPAT;	2004/01/06 11:00
		or ballbearing\$)	US-PGPUB;	
			EPO; JPO;	
			DERWENT	1